

Electrochemical Performance of Deep Discharged $\text{Li}_4\text{Ti}_5\text{O}_{12}$ Anode Material for Lithium-Ion Batteries

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Lithium titanate ($\text{Li}_4\text{Ti}_5\text{O}_{12}$, LTO) is considered a promising anode material for next-generation lithium-ion batteries due to its excellent thermal stability, long cycle life, and intrinsic safety [1]. Its “zero-strain” property during Li^+ intercalation/deintercalation minimizes volume change and ensures outstanding structural stability [2]. The relatively high voltage plateau (~1.55 V vs. Li/Li^+) suppresses lithium dendrite formation, providing superior safety compared with graphite anodes [3]. The spinel structure of LTO offers abundant Li-ion sites, allowing a theoretical capacity of 293 mAh g^{-1} when Ti^{4+} is fully reduced to Ti^{3+} under deep-discharge conditions near 0 V [4]. However, most studies have examined the 1.0–3.0 V range, with limited attention to performance below 1.0 V [4]. To improve conductivity and Li^+ diffusivity, doping at Ti or Li sites and *in situ* or *ex situ* carbon coating are effective strategies that enhance electrochemical performance and rate capability [5].

In the present work, we investigate the electrochemical behavior of Na- and Yb-co-doped, carbon-coated LTO anodes under deep-discharge conditions at room temperature. Pristine $\text{Li}_4\text{Ti}_5\text{O}_{12}$ (LTO) and $\text{Li}_{3.98}\text{Na}_{0.02}\text{Ti}_{4.98}\text{Yb}_{0.02}\text{O}_{12}$ (LTO–Na–Yb), along with their carbon-coated derivatives (LTO@C and LTO–Na–Yb@C), were synthesized via a conventional solid-state reaction method. The carbon-coated variants were further prepared through an *ex situ* pyrolysis process using organic carbon precursors. Galvanostatic charge–discharge tests were conducted within the potential range of 3.0–0.01 V at current densities of 30, 300, and 500 mA g^{-1} , respectively. Structural, morphological, and electrochemical properties were comprehensively characterized using X-ray diffraction (XRD), field-emission scanning electron microscopy (FESEM), galvanostatic cycling, cyclic voltammetry (CV), and electrochemical impedance spectroscopy (EIS).

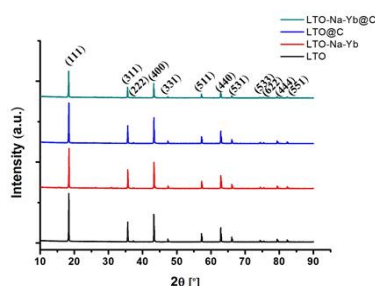


Fig.1. XRD patterns of the pristine LTO and LTO–Na–Yb, LTO@C, LTO–Na–Yb@C.

The phase purity was analyzed by XRD (PANalytical Empyrean, $\text{Cu-K}\alpha$, $2\theta = 10\text{--}90^\circ$). As shown in Fig. 1, both LTO and doped-coated anodes exhibit diffraction peaks

consistent with the cubic spinel structure (JCPDS No. 72-0426). All diffraction peaks correspond to pure cubic LTO, and the refined lattice parameter of pristine LTO and LTO–Na–Yb materials were determined as 8.357 Å and 8.359 Å, respectively.

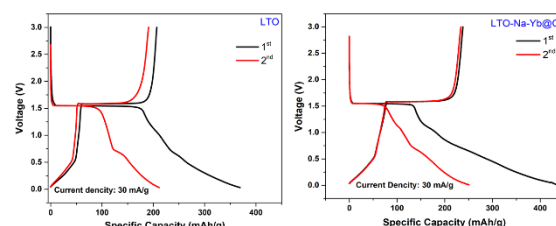


Fig. 2. The charge-discharge profiles of LTO and LTO–Na–Yb@C anodes at 30 mA/g in the range of 0.02–3 V (vs. Li/Li^+)

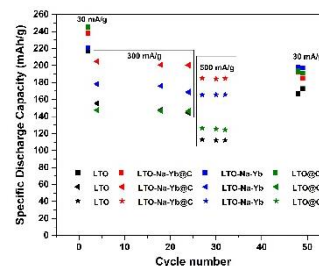


Fig. 3. Rate capabilities of LTO and LTO–Na–Yb, LTO@C, LTO–Na–Yb@C anode materials

As can be seen in Figs. 2 and 3, among all the as-prepared composite anodes, LTO–Na–Yb@C delivers the highest initial discharge capacity across various current densities. Furthermore, it exhibits superior rate capability compared with pristine LTO, LTO–Na–Yb, and LTO@C at 30, 300, and 500 mA g^{-1} .

Acknowledgements

The authors thank the Erciyes University Scientific Research Projects Unit for financial support and ASPİLSAN Energy for their support.

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